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# Rutile TiO<sub>2</sub> films as electron transport layer in inverted organic solar cell

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## Abstract

Titanium dioxide (TiO<sub>2</sub>) thin films were prepared by sol–gel spin coating method and deposited on ITO-coated glass substrates. The effects of different heat treatment annealing temperatures on the phase composition of TiO<sub>2</sub> films and its effect on the optical band gap, morphological, structural as well as using these layers in P3HT:PCBM-based organic solar cell were examined. The results show the presence of rutile phases in the TiO<sub>2</sub> films which were heat-treated for 2h at different temperatures (200°C, 300°C, 400°C, 500°C and 600°C). The optical properties of the TiO<sub>2</sub> films have altered by temperature with a slight decrease in the transmittance intensity in the visible region with increasing the temperature. The optical band gap values were found to be in the range of 3.28-3.59eV for the forbidden direct electronic transition and 3.40-3.79eV for the allowed direct transition. TiO<sub>2</sub> layers were used as electron transport layer in inverted organic solar cells and resulted in a power conversion efficiency of 1.59% with short circuit current density of 6.64mA.cm<sup>-2</sup> for TiO<sub>2</sub> layer heat-treated at 600°C.

## 1. Introduction

In the recent years, significant attention has been paid to the organic solar cells (OSCs) based on bulk-heterojunction (BHJ) materials, due to their low manufacturing cost and potential applications in the field of flexible and large-area solar cells [1]. It could be said that poly(3-hexylthiophene) (P3HT) and (6,6)-phenyl-C61 butyric acid methyl ester (PCBM) are considered as the most investigated materials for OSCs [2]. The inverted organic solar cells (IOSCs) exhibit competitive performance and offer better stability compared with the traditional OSCs [3]. The nature of the contacts on both sides of the active layer is considered as an essential feature to determine the OSCs parameters, hence the final power conversion efficiency (PCE) [4]. Titanium dioxide (TiO<sub>2</sub>) films are used in wide range of applications such as photocatalysis, separations, sensor devices, paints, and solar cells [5]. Mainly, TiO<sub>2</sub> exists in three phases: brookite, anatase and rutile. The Rutile

phase is the most stable phase, however, the solution-based preparation methods for TiO<sub>2</sub> generally shows the anatase phase [6]. Moreover, large anatase particles are difficult to synthesize, because of the transformation from anatase to rutile by increasing treatment times and/or temperature [7]. In an IOSC, a thin film of N-type metal oxide, such as titanium oxide (TiO<sub>2</sub>) is used as a cathode buffer layer in order to lower the work function of the indium tin oxide (ITO) substrate. TiO<sub>2</sub> is an attractive material due to its non-toxicity, low cost, excellent stability, high electron mobility, and high transparency in the visible wavelength range [8]. Chen [9] have reviewed the basic synthetic methods of TiO<sub>2</sub> nanomaterials and films such as sol-gel, hydrothermal, solvothermal and microwave methods. They have demonstrated that the sol-gel method is a versatile method which can produce even amorphous or low crystalline TiO<sub>2</sub> films at low temperatures. Nevertheless, the latter TiO<sub>2</sub> structures could result in low electron mobility when using in the solar cells applications and consequently result in poor device performance [10], thus, the use of heat treatment combined with sol-gel method is desired. There are several techniques that can be used to prepare TiO<sub>2</sub> films such as sol-gel method which is believed to hold the greatest potential for large-scale commercial [11]. PV performance is well-defined by its PCE which is determined by on several factors such as light harvesting and the exciton diffusion length, separation, transportation and collection by the selective electrodes in the solar cell [12]. Though, the adjustment of both contacts has shown to play significant role in the improvement of charge collection [13]. The heat treatment of TiO<sub>2</sub> thin films is known as one of the key factors to adjust the morphology; various TiO<sub>2</sub> nanostructures films have prepared by solution processing methods to acquire efficient charge transport layer [8, 14] as well as changing the structural and optical properties of the thin films [15, 16]. Mainly, Rutile Structure TiO<sub>2</sub> is used in dye syntheses and perovskite solar cells [17-21]. In the current study, the effects of heat treatment on the TiO<sub>2</sub> thin film properties and employing these films as electron transport layer in P3HT:PCBM-based inverted organic solar cells have been investigated. To the best of the authors' knowledge, this is the first attempt to use rutile TiO<sub>2</sub> structure with inverted OSC.

## **2. Samples preparation**

### **2.1. Preparation of TiO<sub>2</sub> films**

Pre-structured ITO coated glass substrates (12Ω/sq sheet resistances) were ultrasonically cleaned in acetone, isopropyl alcohol and deionized water for 10 min each, respectively. TiO<sub>2</sub> films were prepared using sol-gel method as follows: Methanol and isopropanol were

mixed together and used as solvents; Titanium isopropoxyde (TIP) with a purity of 97 % (Aldrich) (titanium precursor) was added to the solvents mixture under stirrer and temperature of 80°C. Afterward, acetic acid (which is act as catalyst, cathode buffer layer to modify the work function of ITO and as the electron transport layer (ETL) on ITO bottom electrode) was dropped wisely to the mixture under stirring. Fig.1 illustrates the steps of preparing Rutile TiO<sub>2</sub>. The latter layer was prepared by using sol-gel and spin coating methods; the TiO<sub>2</sub> thin films were spun with the spin speed of 2000rpm for 30sec, (where the film thickness was estimated by ellipsometry is about 190nm), followed by heat treatment in a programmable furnace at 200 °C, 300 °C, 400 °C, 500 °C and 600 °C.

## 2.2. Preparation of OSCs

Chlorobenzene solution of poly (3- hexylthiophene) (P3HT) (Sigma-Aldrich) and [6,6]-phenyl C61-butyric acid methylester (PCBM) (Sigma-Aldrich) with a mixture ratio of 1:1 [22] was employed to prepare the active layer. This active layer was deposited by spin coating method onto the TiO<sub>2</sub> layer (which was previously prepared onto ITO coated glass substrates), followed by annealing at hot plate inside the nitrogen filled glove box at 120°C for 10min [23]. Hole transport layer consist of Poly(3,4ethylenedioxythiophene): poly(styrenesulfonate) (PEDOT:PSS) (Sigma-Aldrich) (conductive grade) is used; this layer was spun coated onto the active layer. Further heat treatment in the nitrogen-filled glove box at 110 °C was carried out for 10 min. A top contact of gold (Au) was then thermally evaporated with a thickness of 100 nm under vacuum of about 10<sup>-6</sup> mbar, at the deposition rate of 0.1-0.2nm.sec<sup>-1</sup> (with the process monitored by a quartz crystal thickness monitor) onto the PEDOT:PSS layer surface. The schematic diagram of device under study is shown in Fig. 2.

## 2.3. Characterization

The UV-visible absorption spectra of the studied layers were recorded on Varian 50-scan UV– visible spectrophotometer working in the range of 190–1100nm. The blend structure was investigated by multipurpose X'Pert Philips X-ray diffractometer (MPD) (Cu,  $\lambda = 0.154$  nm). The films' morphology was studied using a Nanoscope IIIa multimode atomic force microscope (Bruker-AFM) and FEI-Nova SEM scanning electron microscope. The films' thickness was determined using M2000 (J.A. Woollam Co., Inc.) spectroscopic ellipsometer operating in the wavelength range 370-1000 nm. The photovoltaic (PV) properties were determined using 4200 Keithley semiconductor characterization system

(SCS) and the photocurrent was generated under AM 1.5 solar simulator source of 100 mW.cm<sup>-2</sup>. The Fill factor (FF) and the PCE of the solar cells were evaluated using the following equations [24]:

$$\text{PCE} = \frac{J_{\text{max}}V_{\text{max}}}{P_{\text{in}}} \quad (1)$$

$$\text{FF} = \frac{J_{\text{max}}V_{\text{max}}}{J_{\text{sc}}V_{\text{oc}}} \quad (2)$$

where  $J_{\text{sc}}$  is the short circuit current density (mA.cm<sup>-2</sup>),  $V_{\text{oc}}$  is the open-circuit voltage (V),  $P_{\text{in}}$  is the incident light power and  $J_{\text{max}}$  (mA.cm<sup>-2</sup>) and  $V_{\text{max}}$  (V) are the current density and voltage at the point of maximum power output in the J-V curves [25].

### 3. Results and discussions:

#### 3.1. Optical characteristics

The transmittance spectra of the studied TiO<sub>2</sub> films under different heat-treatment temperature are illustrated in Fig.3. Generally, TiO<sub>2</sub> thin film is highly transparent in the visible range 310–700nm. TiO<sub>2</sub> film which was heat treated at 200°C has shown high transmittance spectra in the range of wavelength from 750-400nm, which is useful for the solar cell applications; transmittance spectra shows a slight decrease by increasing of heat treatment. TiO<sub>2</sub> thin film is well-known as a direct band gap transition semiconductor, plots of the  $h\nu^2$  vs the energy of the absorbed light affords the TiO<sub>2</sub> band gap [26]. The absorption coefficient ( $\alpha$ ) of TiO<sub>2</sub> thin films were calculated from the optical transmittance measurements using the following equation [27]:

$$\alpha = \frac{2.303}{d} \times \log \left( \frac{1}{T} \right) \quad (3)$$

where, T is the transmittance and d is the thickness of the TiO<sub>2</sub> films. The absorption coefficient can also be described as a function of incident photon energy (eV) to calculate the energy band gap ( $E_g$ ) for the studied films using the following relation:

$$\alpha h\nu = A(h\nu - E_g)^n \quad (4)$$

where, h is the Planck's constant,  $\nu$  is frequency, A is constant and n is a value depends on the nature of transition. Generally, the rutile TiO<sub>2</sub> belongs to direct band gap semiconductor category [28]. Therefore, when n= 1/2 the band gap transition is allowed direct and the extrapolation of the linear region of  $(\alpha h\nu)^2$  vs.  $h\nu$  curve at  $\alpha = 0$  provides the direct energy band gap of TiO<sub>2</sub> thin film [27]; whereas n= 3/2 the band gap transition is forbidden direct. Moreover, the inset of Fig.3 shows the absorption spectra of the P3HT:PCBM active layer which shows four bands. Bands A, B and C were correlated to

the absorption of the P3HT main polymer while the band D is related to the absorption in the PCBM domains [29]. Fig.4 shows Tauc plot to determine the energy band gap for the studied TiO<sub>2</sub> films. The energy gap values depend in general on the films crystal structure, and the arrangement and distribution of the atoms in the crystal lattice. Also it is affected by crystal regularity [7]. The allowed direct band gap (Fig.4A) was found to be increase with increasing the temperature from 3.4eV (200°C) to 3.79eV (600°C). On the other hand, the forbidden direct band gap (Fig.4B) has increased from 3.28eV (200°C) to 3.59eV (600°C); the results are shown in table 1. Usually rutile phase has a band gap higher than 3eV, however the differences in our results have been correlated to the structure properties which lead to different electronic configuration; where the Ti-Ti and Ti-O distances in TiO<sub>2</sub> structure are different between anatase and rutile phase [30].

### 3.2. Structural properties

The structural properties of the TiO<sub>2</sub> film were analyzed by X-ray diffraction as shown in Fig.5. All the TiO<sub>2</sub> films have exhibited polycrystalline structure; the TiO<sub>2</sub> structure shows no anatase phases, whereas clear peaks of Rutile phases have been observed at around 27° (110), 34° (101), 38° (200), 51° (211), 61° (002) and 66° (310) [31, 32]. The fine variation in the temperature has shown a slight increase in the intensity of XRD peaks and better crystallinity take place at higher temperature (600°C). In the XRD patterns, the (110) reflection appears anomalously intense indicating that the material is preferentially oriented [7]. Other diffraction peaks were observed at different phases, these are, (101, 200, 211, 002 and 310) which were attributed to the rutile phases of the TiO<sub>2</sub> structure [7]. To determine the structure parameters, the main XRD peaks at 2Theta were fitted with one Gaussian to obtain the peaks' width and the lattice constants as illustrated in Table 2. Moreover, the grain sizes (D) of the nanocrystalline structure for the studied TiO<sub>2</sub> layers were calculated using Scherrer equation [29]:

$$D = \frac{k\lambda}{\beta \cos(\theta)} \quad (5)$$

where D is the grain size,  $\lambda$  is the wavelength of the radiation, k is called Scherrer's constant which is equal to 0.94,  $\beta$  is the peak width at half-maximum intensity and  $\theta$  is the peak position. From table 2, it is obvious that the crystal size of the TiO<sub>2</sub> has increase upon increasing the temperature resulting in higher grain size for the main two peaks of the rutile structure (110 and 200).

### 3.3. Morphological properties

The morphological properties of the studied TiO<sub>2</sub> layer treated at different temperatures were analyzed using SEM and AFM techniques. Fig.6 shows the SEM images for the TiO<sub>2</sub> films with different heat-treatment temperatures. TiO<sub>2</sub> surface morphology has exhibited a clear porous nature; this morphology has changed upon increasing the heat-treatment temperature to 500°C and 600°C to become bigger grain size and more homogeneous surface morphology. The film heat treatment at 300, 400 and 500 °C showed some granular structures with irregular sizes. At 600°C heat treatment, the films exhibit irregular and grainier structure with no clear pinholes, which is in good agreement with XRD results for increasing the grain size with increasing temperature. The presence of pin holes usually decreases the efficiency of solar cells by shorting the back contact to the front Transparent Conducting Oxide (TCO) layer. These results confirm that heat treatment enhances the recrystallization and the grains became denser to fill the gaps and form a uniform thin layer of TiO<sub>2</sub>. Similar results have been observed from the AFM images as shown in Fig.7. The topographic images have demonstrated that the grain size increases with the increasing of heat treatment temperature; the surface roughness (RMS) of the TiO<sub>2</sub> films was estimated for the studied films as (0.129nm ,0.273nm, 0.512nm, 0.610nm and 0.920nm) for the heat treatment temperatures 200°C, 300 °C, 400 °C, 500 °C and 600 °C, respectively.

### 3.4. Photovoltaic characteristics

The photovoltaic properties of the inverted organic solar cells based on P3HT:PCBM active layer using TiO<sub>2</sub> as electron transport layer was investigated. Fig.8 and table 3 show J-V characteristics of the (ITO/TiO<sub>2</sub>/P3HT:PCBM/PEDOT:PSS/Au) solar cells with different heat treatment of TiO<sub>2</sub> from (200-600 °C). Devices with TiO<sub>2</sub> treated at higher temperature have exhibited an increase in the short circuit density current ( $J_{sc}$ ) from 1.06 mA.cm<sup>-2</sup> to 6.64 mA.cm<sup>-2</sup> and a slight increase in the open circuit voltage ( $V_{oc}$ ) from 0.59V to 0.60V. The corresponding results show that the obtained solar cell with the TiO<sub>2</sub> thin film heat treatment of 600 °C has exhibited better photovoltaic properties, indicating that more electrons can be extracted and transported to ITO [33]. Moreover, PCE of the studied devices have revealed a clear increase in PCE from 0.35% in the device with TiO<sub>2</sub> treated at 200°C to 1.6% in the device with TiO<sub>2</sub> treated at 600°C. However, the FF has shown a clear reduction from 55.5% to 42.3% upon increasing the TiO<sub>2</sub> heat treatment temperature.

This may be attributed to the increase in the series resistance [23]. The series resistance of the studied devices has shown an increase from  $9.6(\Omega\cdot\text{cm}^{-2})$  in the device with  $\text{TiO}_2$  treated with temperature at  $200^\circ\text{C}$  to  $48.9(\Omega\cdot\text{cm}^{-2})$  in the device treated at  $600^\circ\text{C}$ . However, PCE for devices with an active area less than  $0.1\text{cm}^2$  show slight sensitivity to the cathode conductivity [34]. Generally, two basic processes are controlled the J-V characteristics of OSCs: carrier injection in the bulk material by the electrodes and charge transport in the bulk material [35]. The carriers' injection process is also governed by different processes: Schottky emission (thermionic emission), Poole–Frenkel emission (field-enhanced thermal excitation of trapped charges) and charge tunneling (Fowler–Nordheim tunneling from the metal contact into the HOMO and LUMO levels of the polymer) [36]. The injected carriers by the electrode have to overcome Schottky barrier or tunnel through it. After the carriers' injection, the conduction properties of the active layer play the main role in transportation toward the opposite electrode. Mainly, Ohmic conduction is dominated in the charge carriers' transportation in a bulk materials such as P3HT:PCBM at low voltages [37]. This could be attributed to the thermally excited carriers' hopping between the isolated states and the space-charge-limited current (SCLC) is mostly due to carriers' injection from the electrode to the active layer [38]. In the case of P3HT:PCBM-based organic solar cells, an interpenetrated percolation pathways is form to transport the charge carrier where the holes are transported through P3HT domains and electrons through the PCBM domains, which is limited by the charge carriers' mobility [39]. Low charge carrier mobility in organic materials is mostly a function of “mobility gaps” due to the structural disorder [40]. Therefore, recombination is very likely to occur during the transport process, the total resistance will increase exponentially, which is due to the recombination effect [33], which in turn affects the FF.

### **Conclusion:**

Titanium dioxide thin films were prepared using sol-gel process. XRD peaks were observed at different phases, these are, (101, 200, 211, 002 and 310) which were attributed to the rutile phases of the  $\text{TiO}_2$  structure. The latter structure belongs to direct band gap semiconductor category. Using Tauc equation, the allowed direct band gap was found to be increase with increasing the temperature from  $3.4\text{eV}$  ( $200^\circ\text{C}$ ) to  $3.79\text{eV}$  ( $600^\circ\text{C}$ ) and the forbidden direct band gap has increased from  $3.28\text{eV}$  ( $200^\circ\text{C}$ ) to  $3.59\text{eV}$  ( $600^\circ\text{C}$ ).  $\text{TiO}_2$  surface morphology has exhibited a clear porous surface which has changed upon increasing the heat-treatment temperature to  $500^\circ\text{C}$  and  $600^\circ\text{C}$  to become bigger grain size

and more homogeneous surface morphology. The PV properties have shown an increase after the thermal treatment of TiO<sub>2</sub>, FF has shown a clear reduction from 55.5% to 42.3% which could be due to the increase in the series resistance. PCE of 1.59% with high photo current density of 6.64mA.cm<sup>-2</sup> using TiO<sub>2</sub> layer heat-treated at 600°C has been achieved.

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Table.1: The obtained band gap values are given in Table.

<b>TiO<sub>2</sub> heat treatment (°C)</b>	<b>Allowed direct transition Eg(A) (eV)</b>	<b>Forbidden direct transition Eg(F) (eV)</b>
<b>200</b>	3.40	3.28
<b>300</b>	3.57	3.33
<b>400</b>	3.60	3.40
<b>500</b>	3.69	3.44
<b>600</b>	3.79	3.59

Table 2: XRD parameters for TiO<sub>2</sub> films fitted for two peaks (110 and 200).

<b>Temp. (°C)</b>	<b>Phase</b>	<b><math>\theta</math> (°)</b>	<b>d-space (nm)</b>	<b>FWHM (2<math>\theta</math>)</b>	<b>D (nm)</b>
<b>200</b>	110	26.6	3.24	0.2788	39.9
	200	37.9	2.36	0.3134	26.5
<b>300</b>	110	26.6	3.22	0.2764	39.5
	200	37.8	2.37	0.2564	32.4
<b>400</b>	110	26.6	3.23	0.2623	41.9
	200	37.9	2.37	0.2289	36.3
<b>500</b>	110	26.6	3.22	0.258	42.5
	200	37.8	2.37	0.2677	31.0
<b>600</b>	110	26.6	3.22	0.2311	49.2
	200	37.9	2.36	0.2145	38.9

Table 3: The PV parameters for the devices with TiO<sub>2</sub> treated at different temperatures

TiO <sub>2</sub> heat treatment (°C)	V <sub>oc</sub> (volt)	J <sub>sc</sub> (mA.cm <sup>-2</sup> )	FF (%)	PCE	R <sub>s</sub> (Ω.cm <sup>-2</sup> )
200	0.59	1.06	55.5	0.35	9.6
300	0.58	2.06	55.5	0.66	20.6
400	0.58	2.43	55.1	0.78	38.9
500	0.59	3.75	51.2	1.15	41.1
600	0.60	6.64	42.3	1.6	48.9

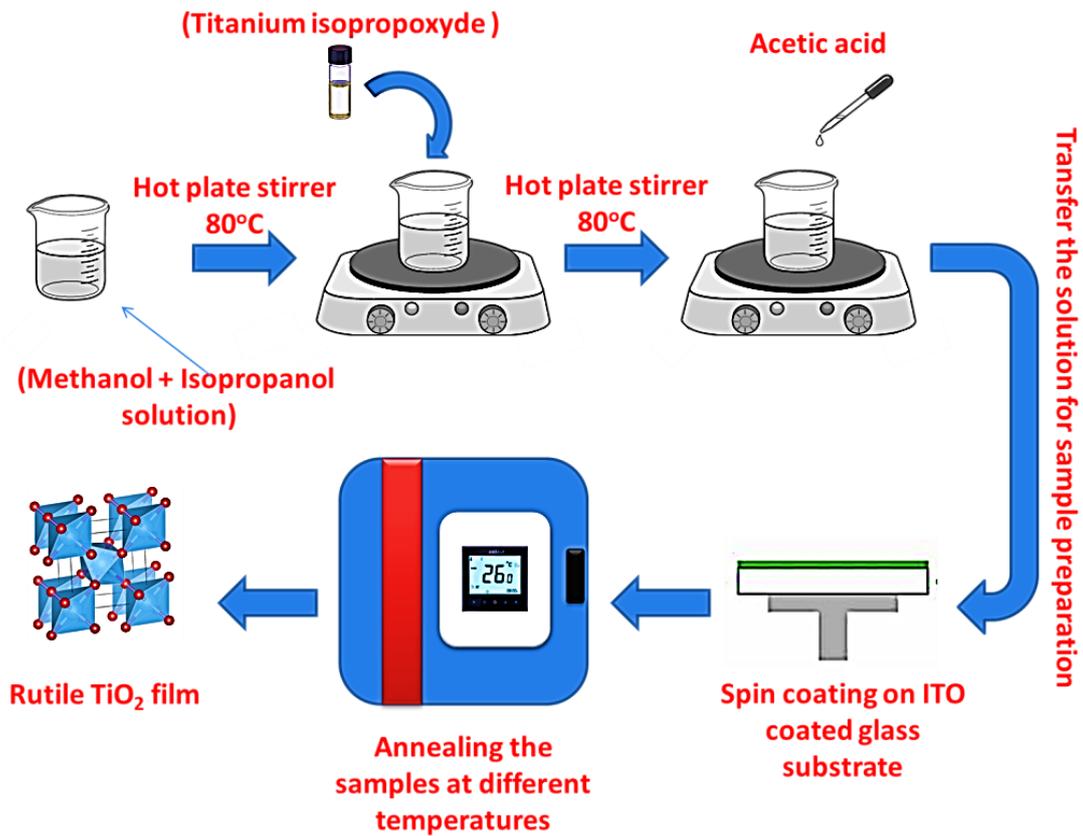


Fig.1: The procedure of preparing Rutile TiO<sub>2</sub>.

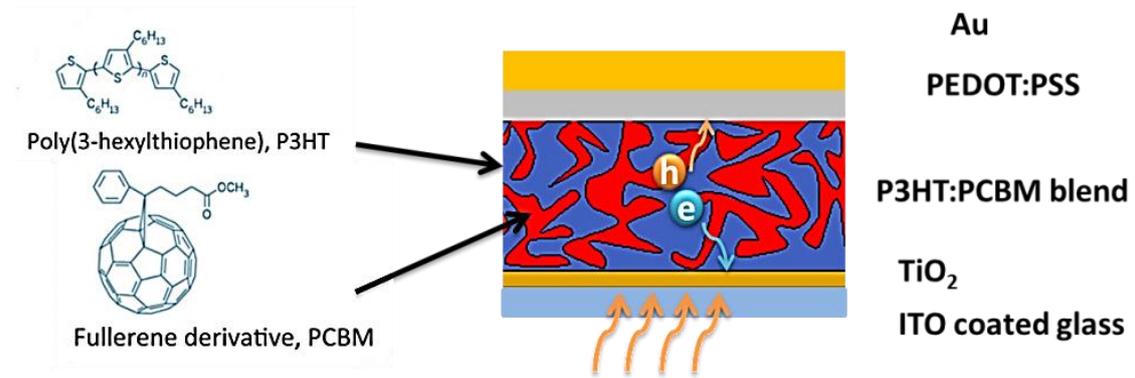


Fig.2: The schematic diagram of devices under study  
 (ITO\TiO<sub>2</sub>\P3HT:PCBM\PEDOT:PSS\Au)

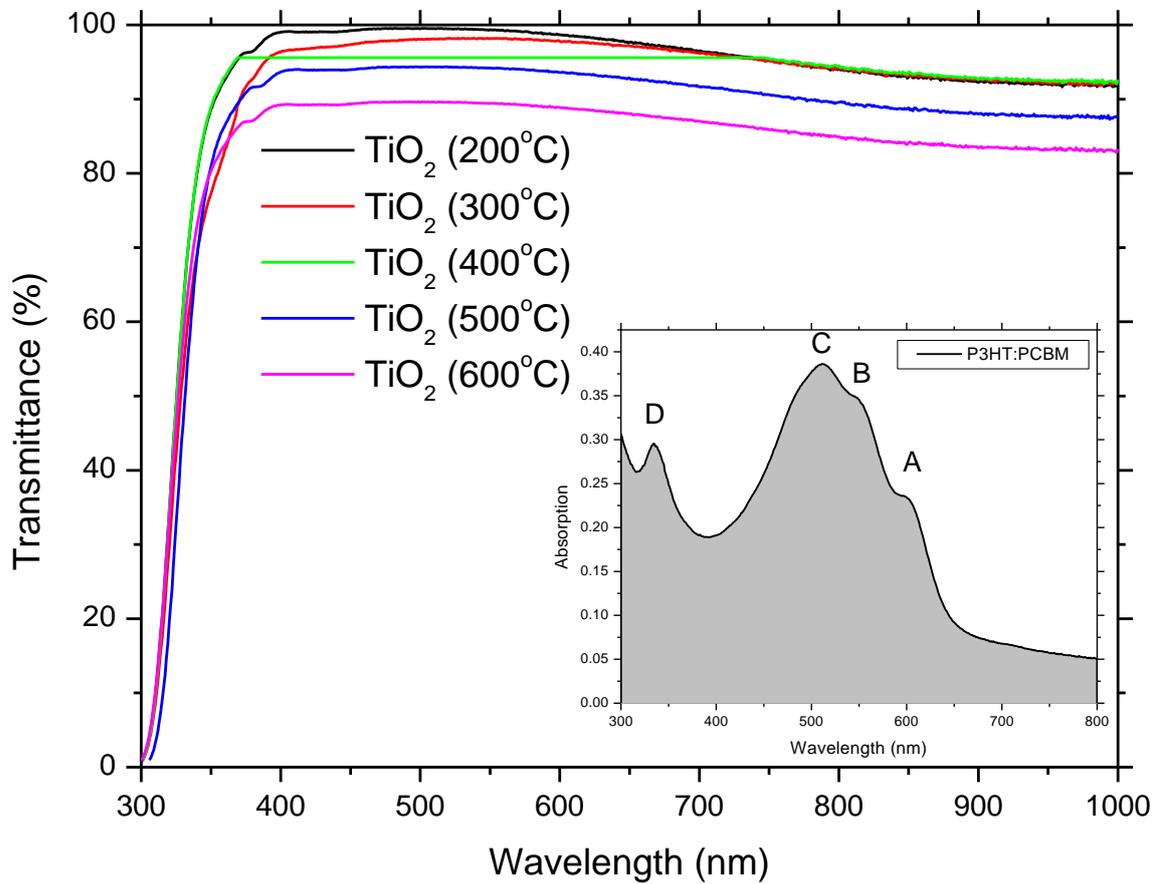


Fig.3: Transmission spectra of TiO<sub>2</sub> films with different heat treatment temperatures; the inset represents the absorption spectra of the P3HT:PCBM active layer.

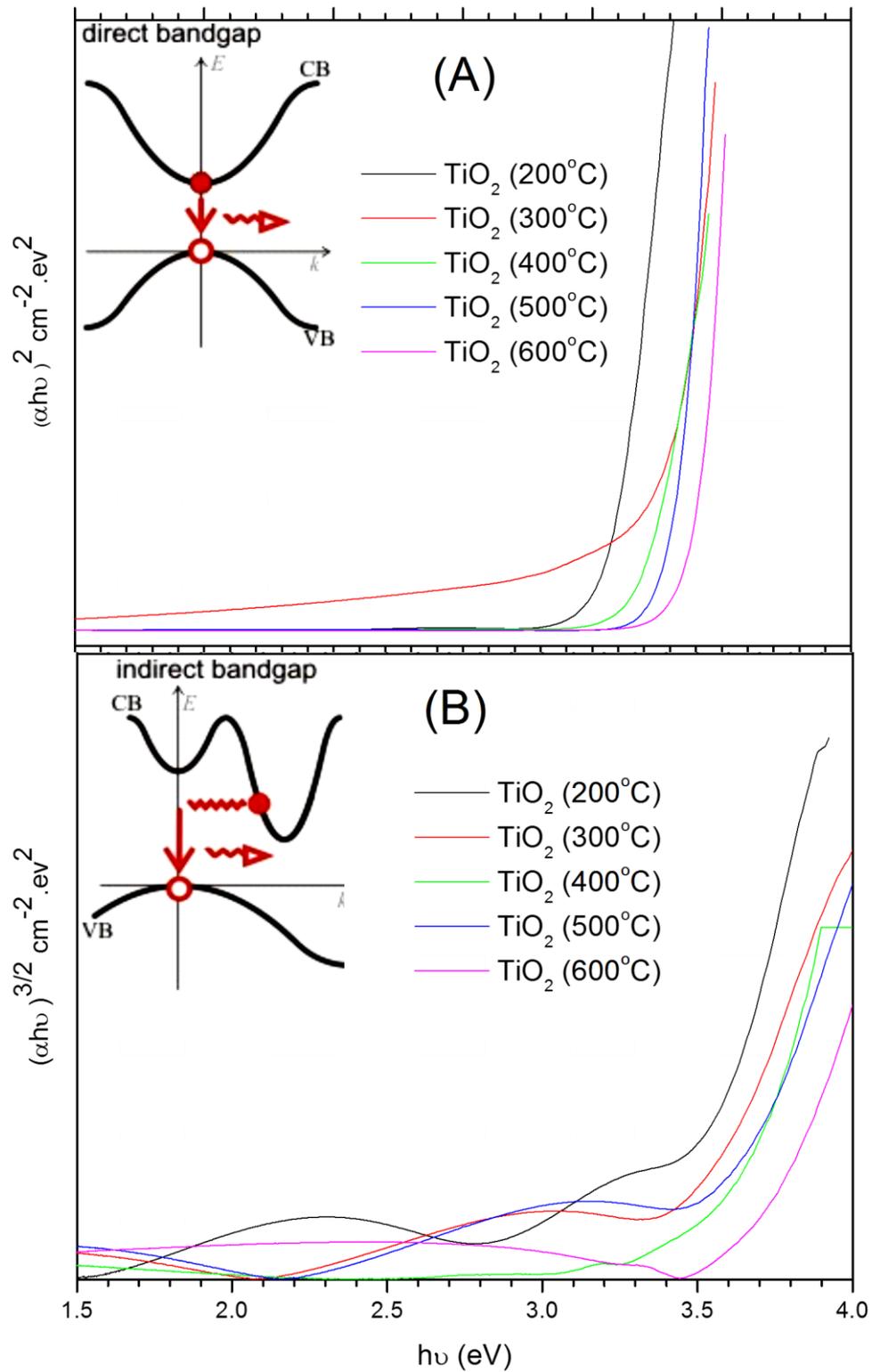


Fig.4: (A) Tauc plot for the TiO<sub>2</sub> films treated at different temperatures (A) direct allowed and (B) Direct forbidden.

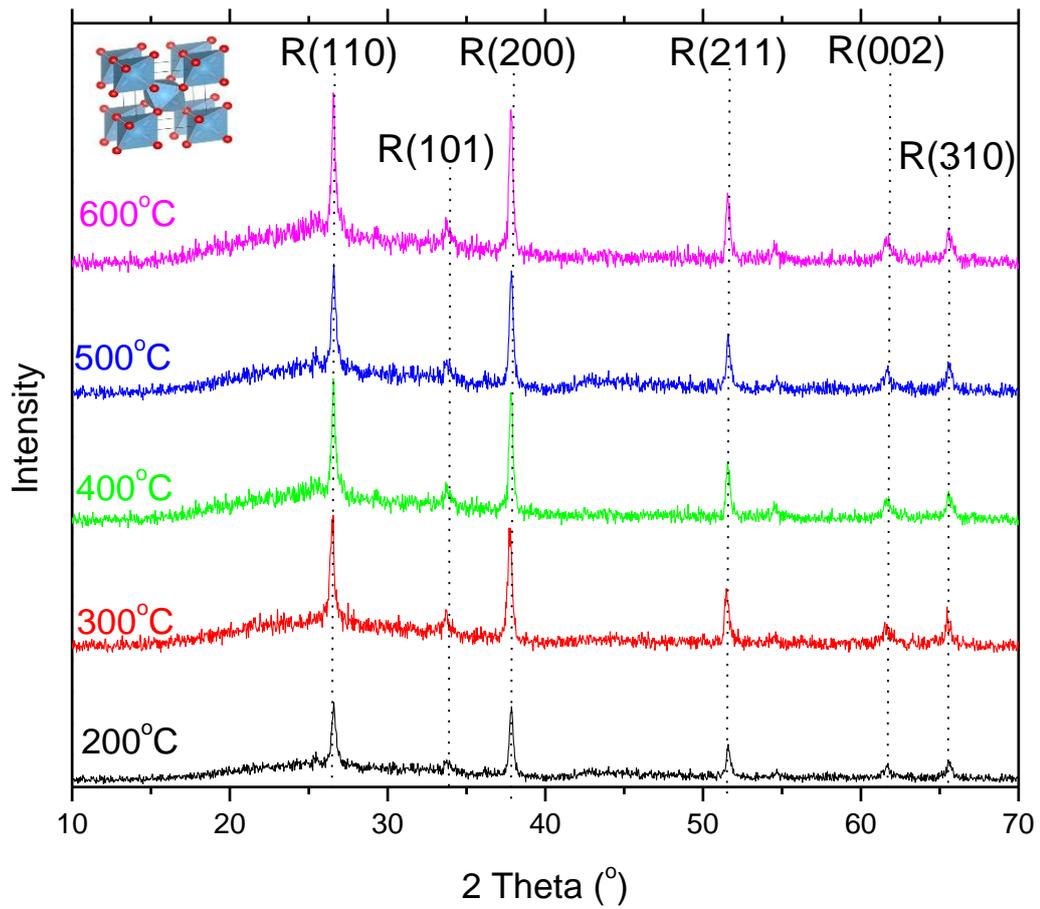


Fig.5: XRD patterns of TiO<sub>2</sub> films grown on ITO- glass substrates at heat treatment temperature 600°C.

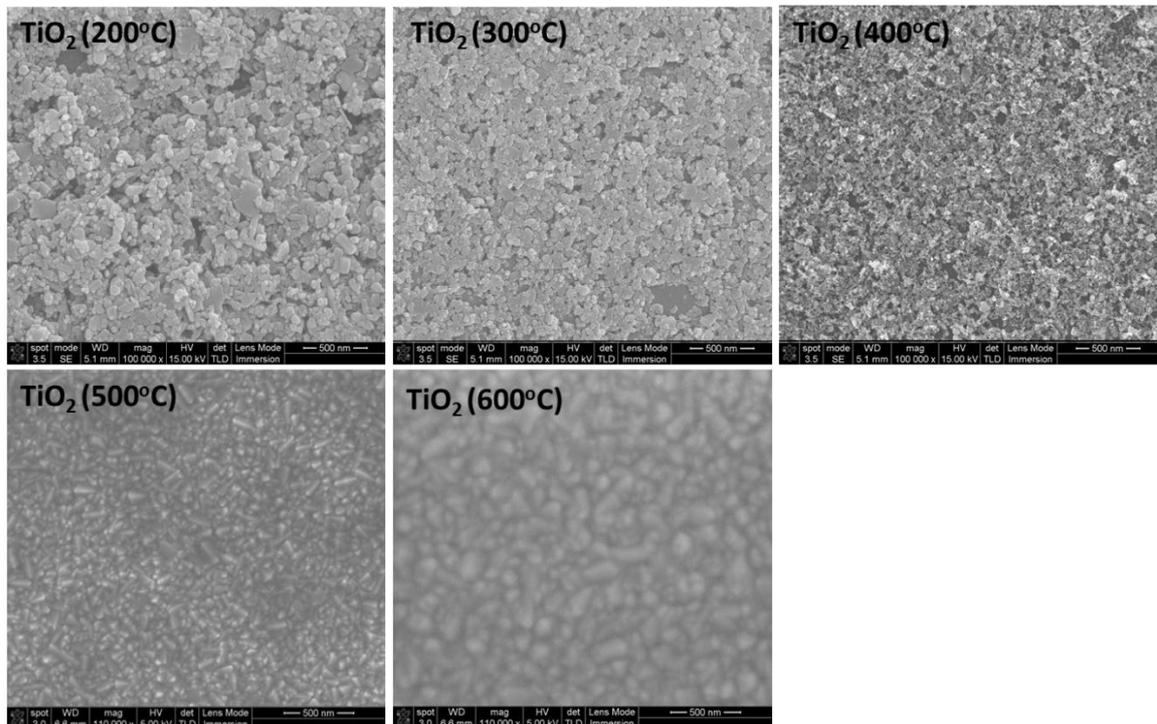


Fig.6: SEM image of TiO<sub>2</sub> films grown on ITO-coated glass substrates at different heat - treatment temperature (200-600°C).

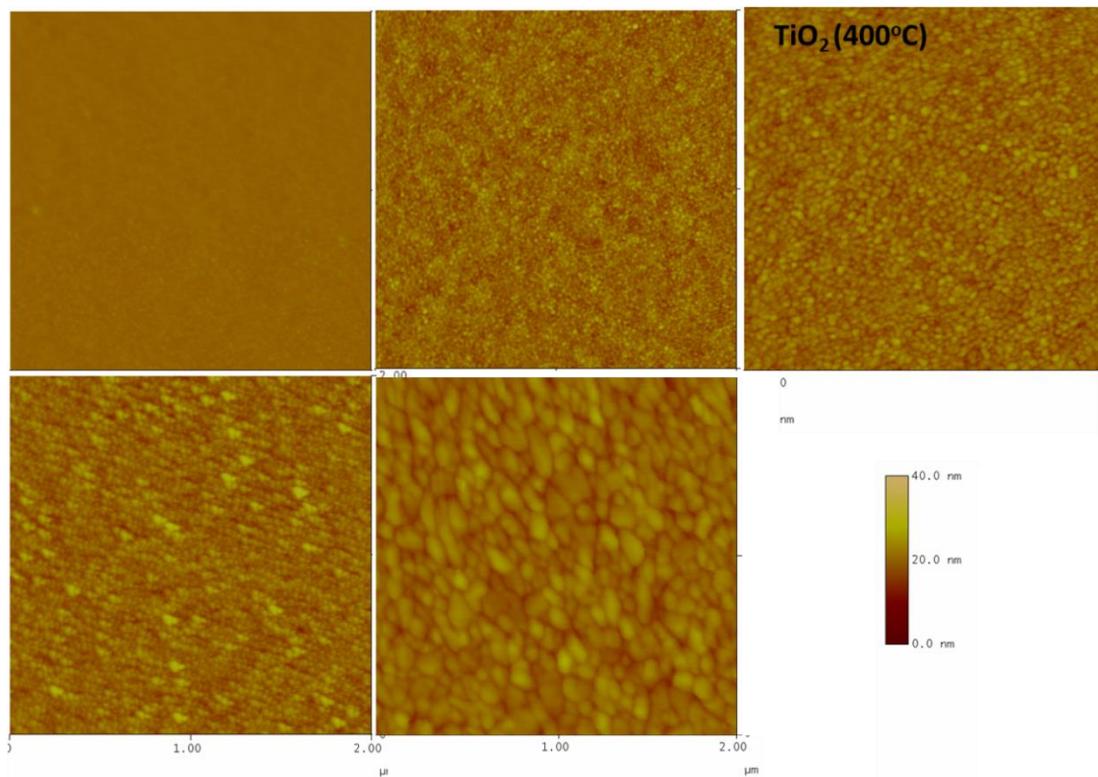


Fig.7: 2D AFM images of TiO<sub>2</sub> films at different heat-treatment temperature (200-600°C)

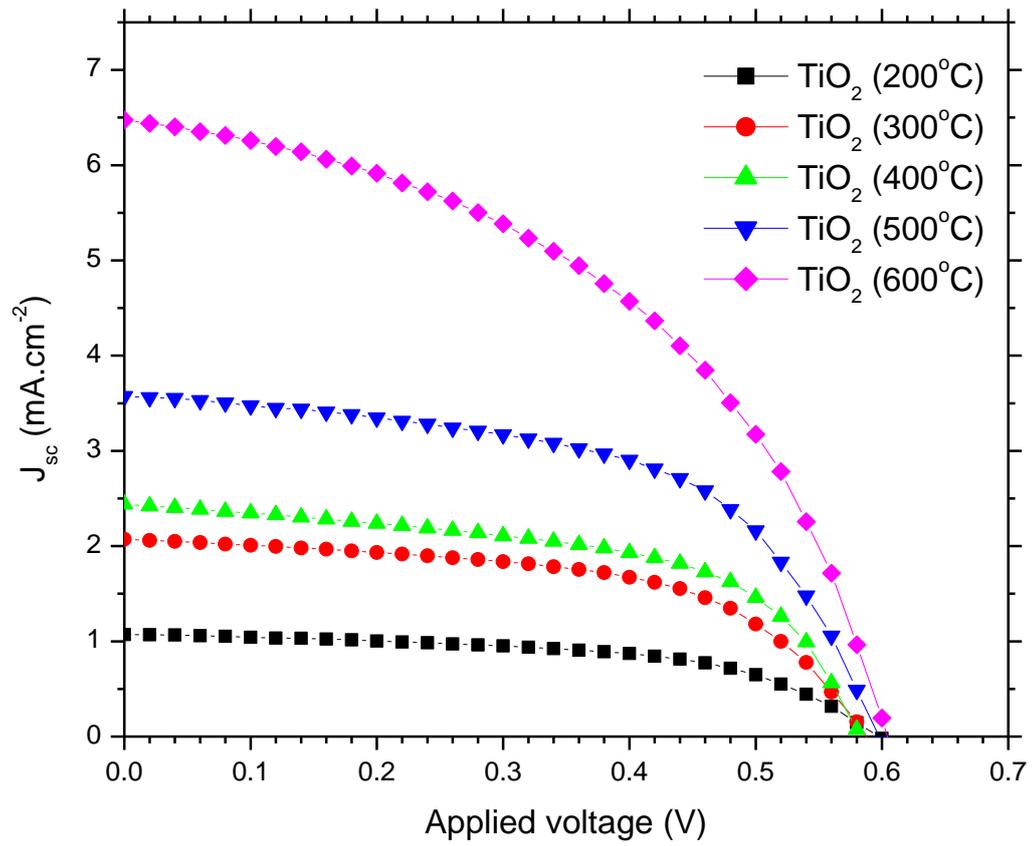


Fig.8: The J–V characteristics of the fabricated inverted solar cells with different heat treatment TiO<sub>2</sub>.